

Kinetic analysis of epoxy-based composite curing

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Resin curing

Kinetic analysis

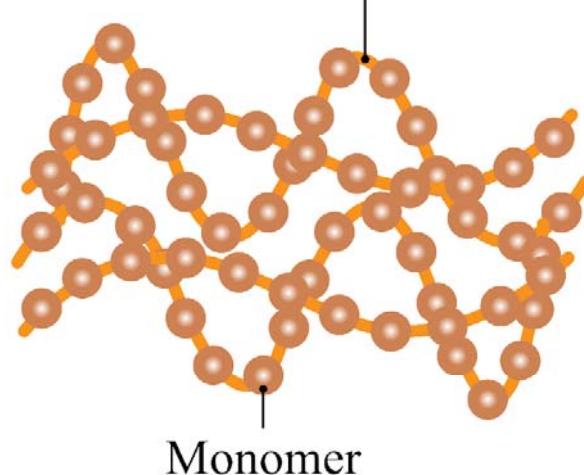
Prediction of the evolution of two epoxy resin
cure

Overheating

Conclusions

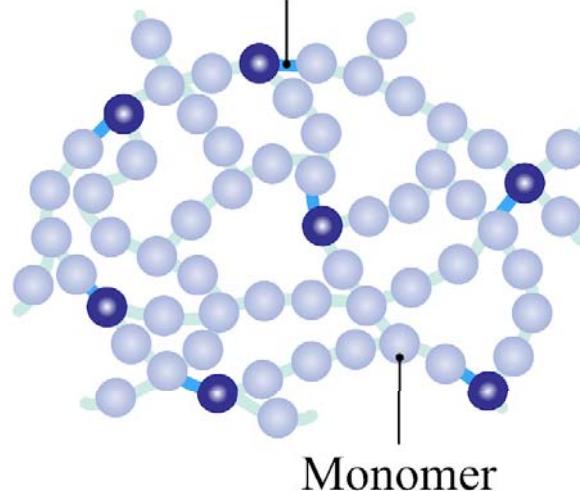
thermoplastic

Strong link into polymer chains



thermoset

Strong cross-link bond



Weak intermolecular forces
between polymer chains

No cross-links between chains

Softens when heated

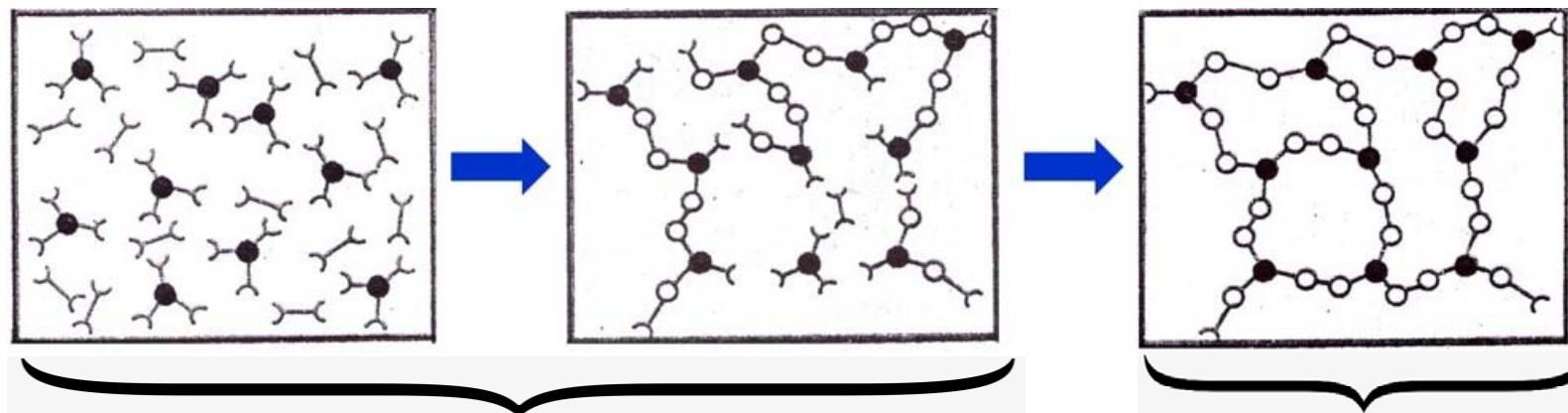
Strong covalent bonds
between polymer chains

Remains hard when heated

Resin curing: formation of three-dimensional cross-linked
thermoset structure

Stages:

- **chain extension:** primary amine reaction.
- **crosslinking:** secondary amine reaction

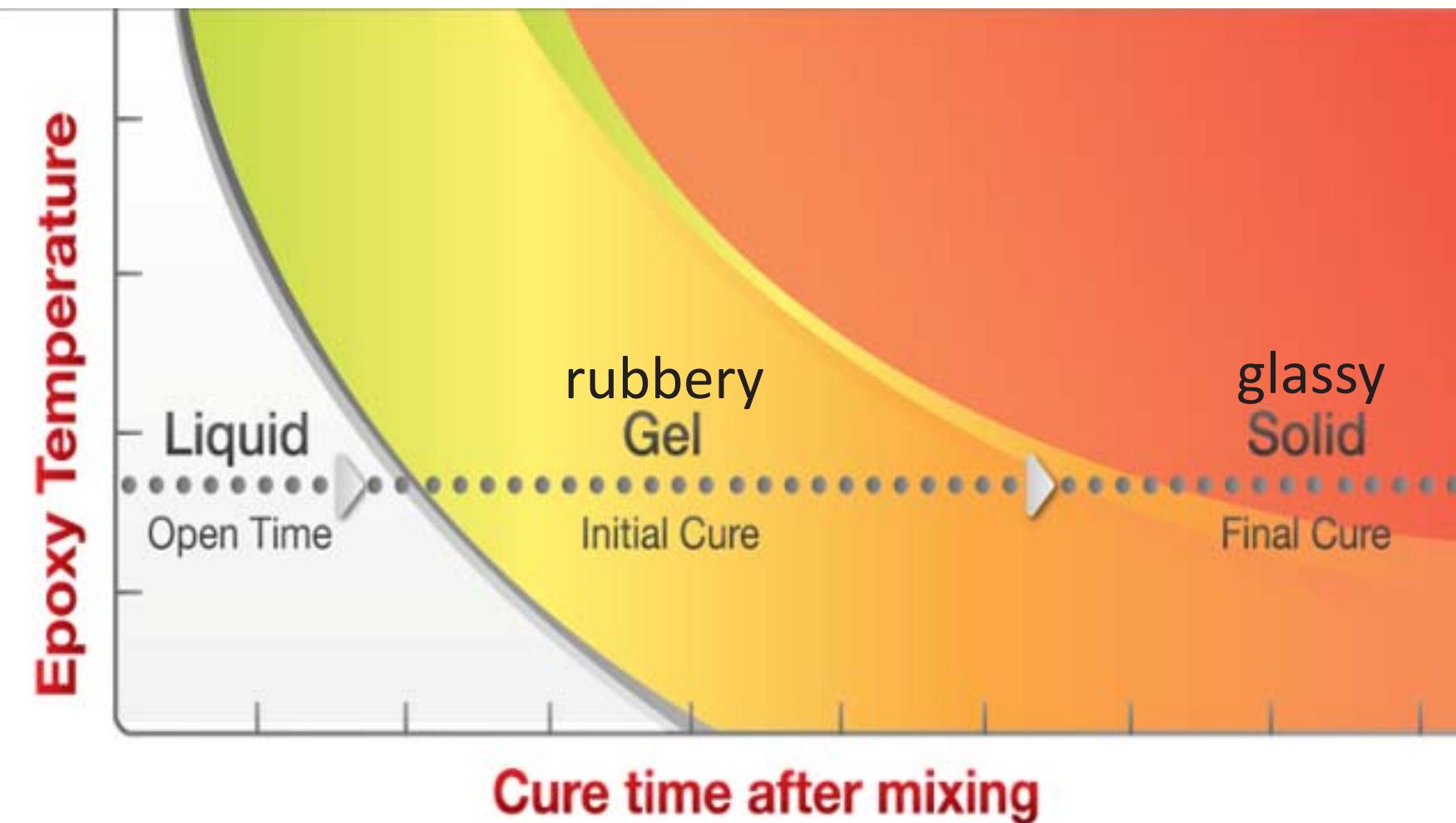


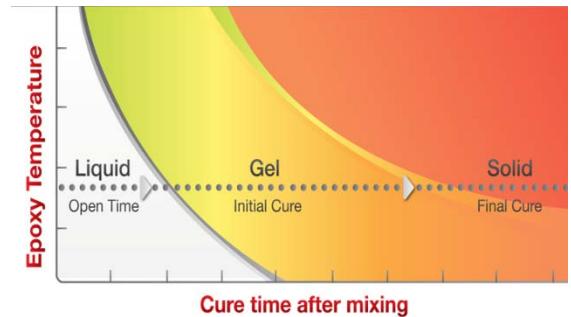
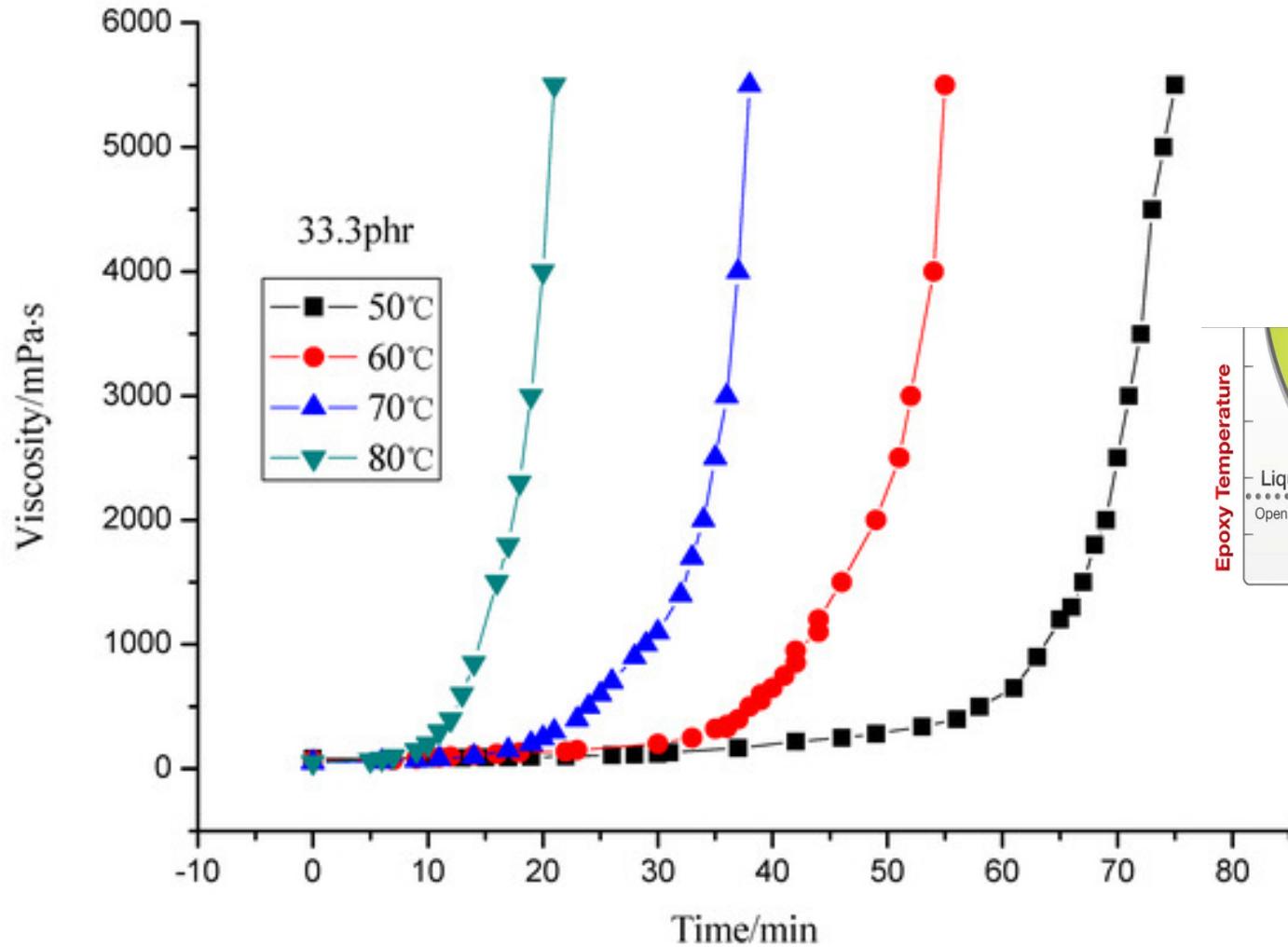
liquid composed of comonomers
and just formed oligomers

rubbery or
glassy solid

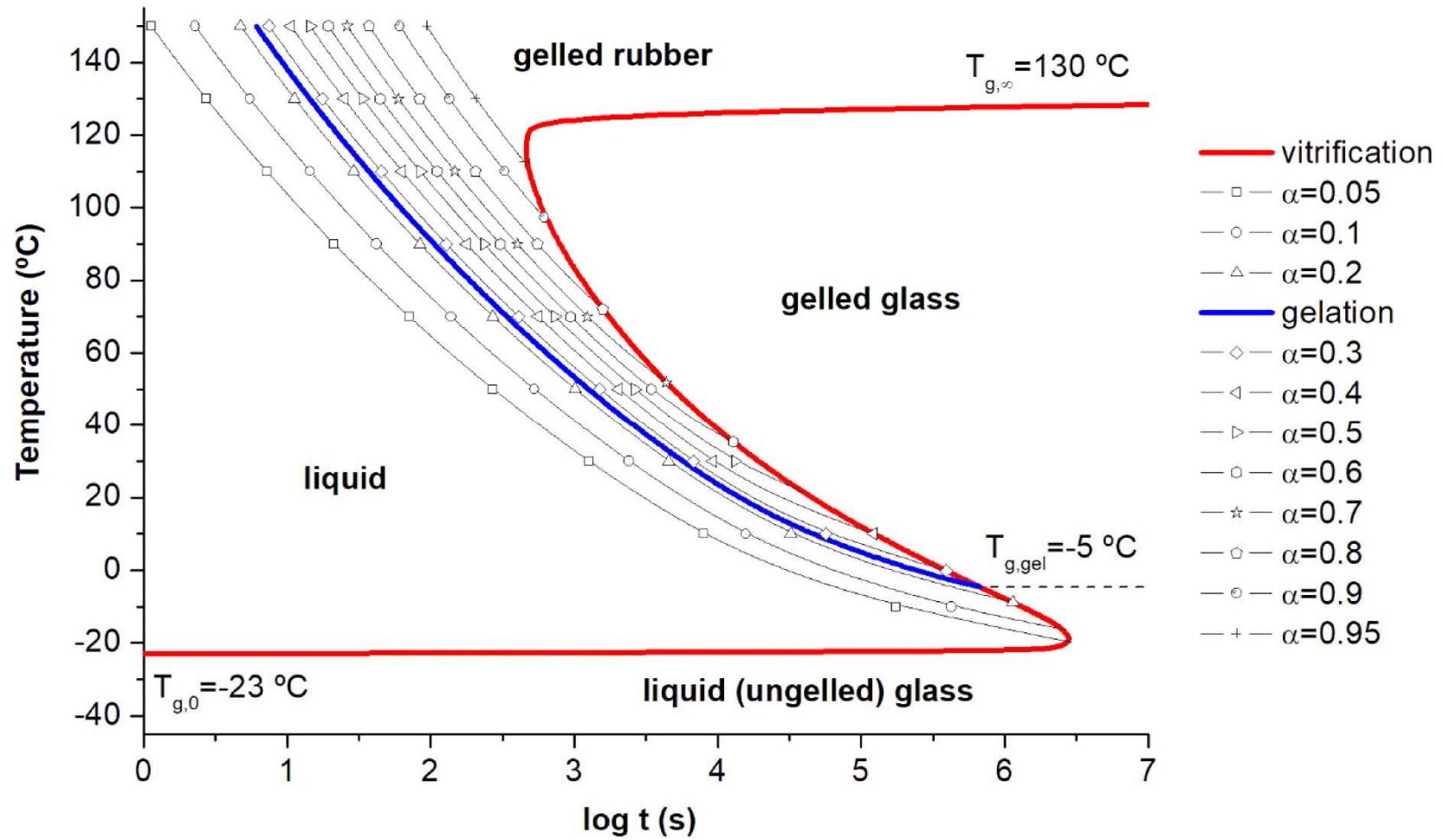
Molecular weight
viscosity
glass transition temperature

time
↑
molecular mobility
↓

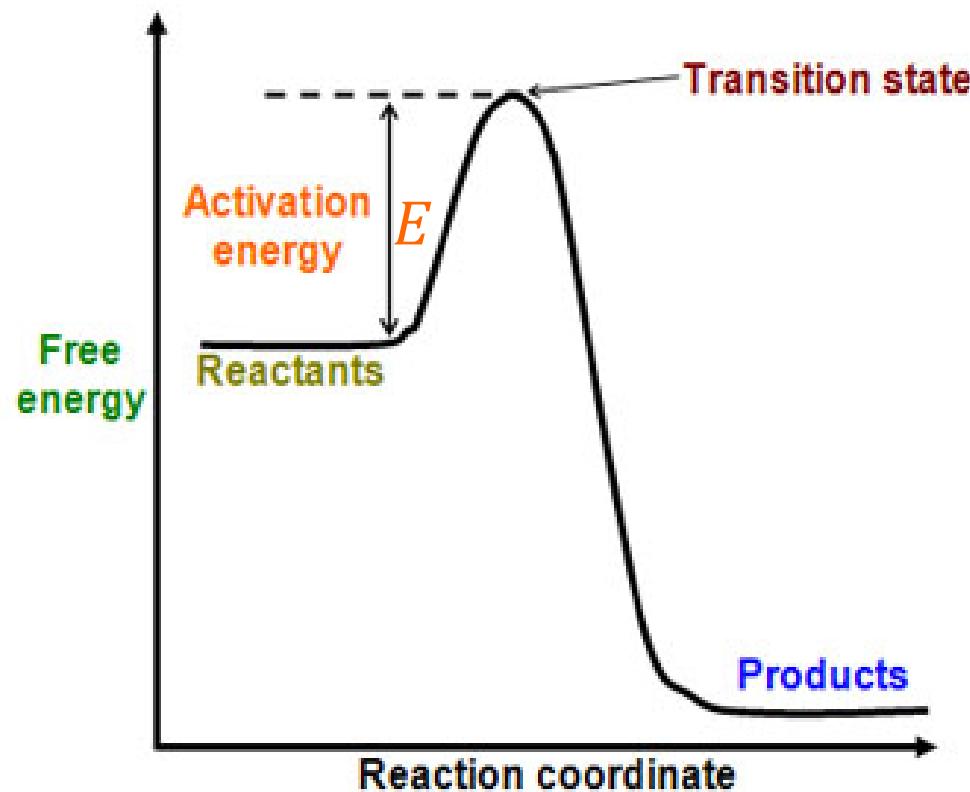




Time-temperature-transformation diagram of a thermosetting system



Thermally activated process



Degree of cure (0-1): α

Curing rate (s^{-1}):

$$\frac{d\alpha}{dt} = k(T)f(\alpha)$$

Rate constant (s^{-1}):

$$k(T) = Ae^{-E/RT}$$

Activation energy (kJ/mol): E

Phenomenological models

- nth order reaction:

$$\frac{d\alpha}{dt} = k(T)(1 - \alpha)^n$$

- Sestak-Berggren (autocatalytic behavior)

$$\frac{d\alpha}{dt} = k(T)\alpha^m(1 - \alpha)^n$$

- Kamal model ($k_1(T)$ non-catalyzed and $k_2(T)$ catalyzed reaction).

$$\frac{d\alpha}{dt} = k_1(T)(1 - \alpha)^n + k_2(T)\alpha^m(1 - \alpha)^n$$

Phenomenological models (diffusion limitations)

$$\frac{d\alpha}{dt} = k(T) \left(\frac{\alpha}{\alpha_{max}(T)} \right)^m \left(1 - \frac{\alpha}{\alpha_{max}(T)} \right)^n$$

maximum fractional conversion at the specific temperature, α_{max} :

$$\alpha_{max}(T) = \frac{T_{g0} T_{g^\infty}}{(T_{g0} - T_{g^\infty})T} - \frac{T_{g^\infty}}{(T_{g0} - T_{g^\infty})}$$

A second approach assumes a serial combination of chemical reaction and diffusion

$$\frac{1}{k(T)} = \frac{1}{k_{diff}(T)} + \frac{1}{k_{chem}(T)}$$

- Model-fitting (phenomenological models)
- Model free methods

Isoconversional methods

$$\left[\frac{d \ln(d\alpha/dt)}{dT^{-1}} \right]_{\alpha} = -\frac{E_{\alpha}}{R} \quad (1)$$

E_{α} is the apparent activation energy

Integration of (1) yields:

$$\frac{d\alpha}{dt} = Af(\alpha)e^{-E_{\alpha}/RT}$$

- Model-fitting (phenomenological models)
- Model free methods

Isoconversional methods

$$\left[\frac{d \ln(d\alpha/dt)}{dT^{-1}} \right]_{\alpha} = -\frac{E_{\alpha}}{R} \quad (1)$$

E_{α} is the apparent activation energy

$$\frac{1}{k(T)} = \frac{1}{k_{diff}(T)} + \frac{1}{k_{chem}(T)}$$

$$E_{\alpha} = \frac{A_{chem} \exp\left(-\frac{E_{chem}}{RT}\right) E_{diff} + A_{diff} \exp\left(-\frac{E_{diff}}{RT}\right) E_{chem}}{A_{chem} \exp\left(-\frac{E_{chem}}{RT}\right) + A_{diff} \exp\left(-\frac{E_{diff}}{RT}\right)}$$

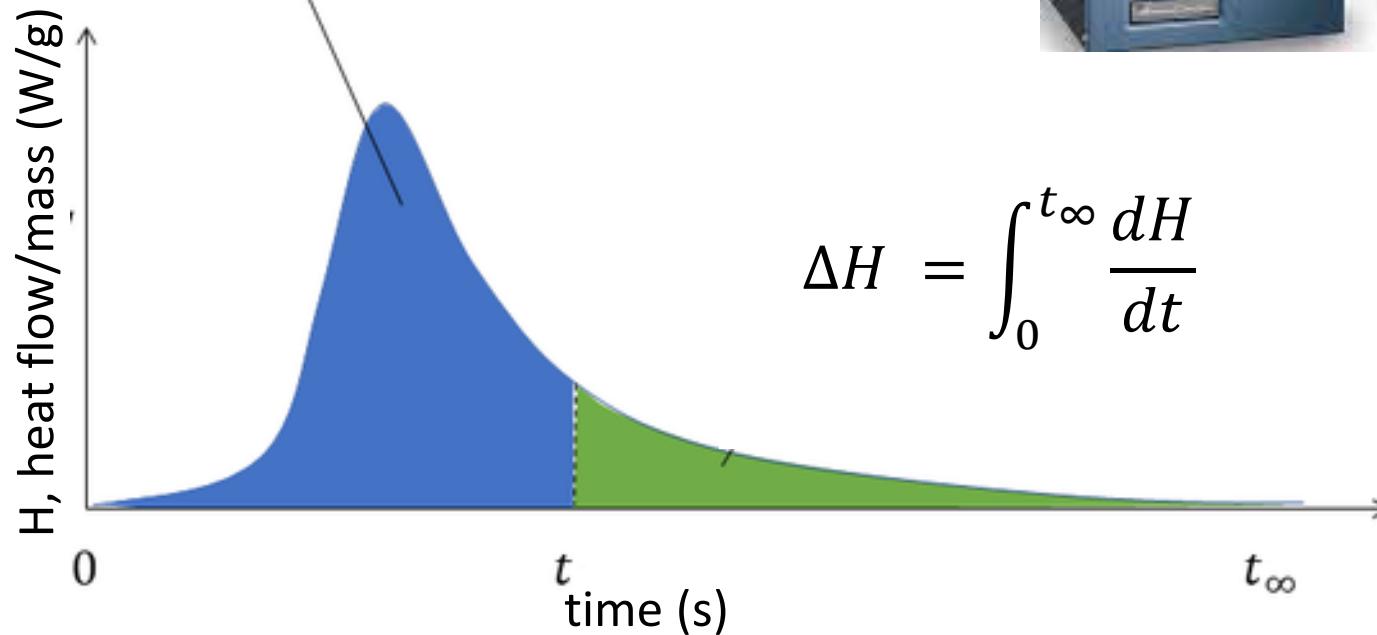
Monitoring methods:

- Rheological analysis (Rheometer, DMA)
- Thermal analysis (DSC)
- Spectroscopic analysis (FTIR, Raman, fluorescence)
- Ultrasonic analysis

- Thermal analysis (DSC)

$$\frac{d\alpha}{dt} = \frac{1}{\Delta H} \frac{dH}{dt}$$

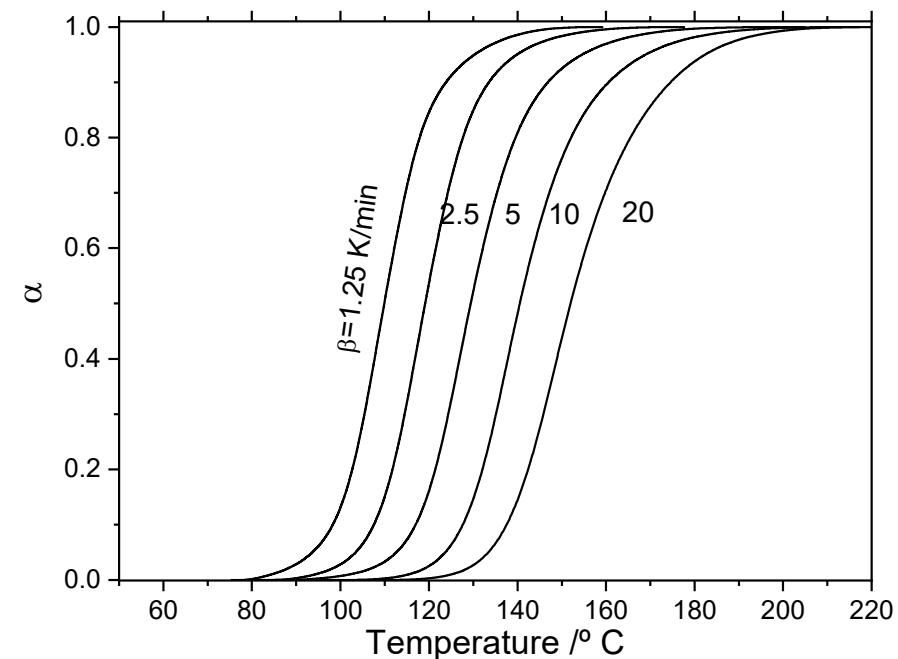
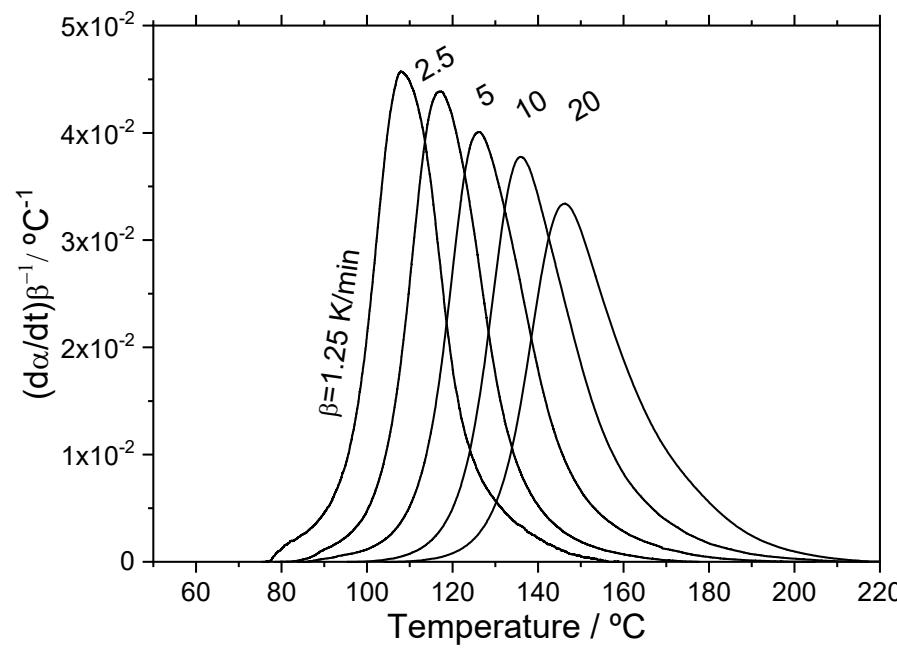
$$\alpha = \frac{1}{\Delta H} \int_0^t \frac{dH}{dt} dt$$



$$\Delta H = \int_0^{t_\infty} \frac{dH}{dt}$$

Temperature dependence?

- Isothermal experiments

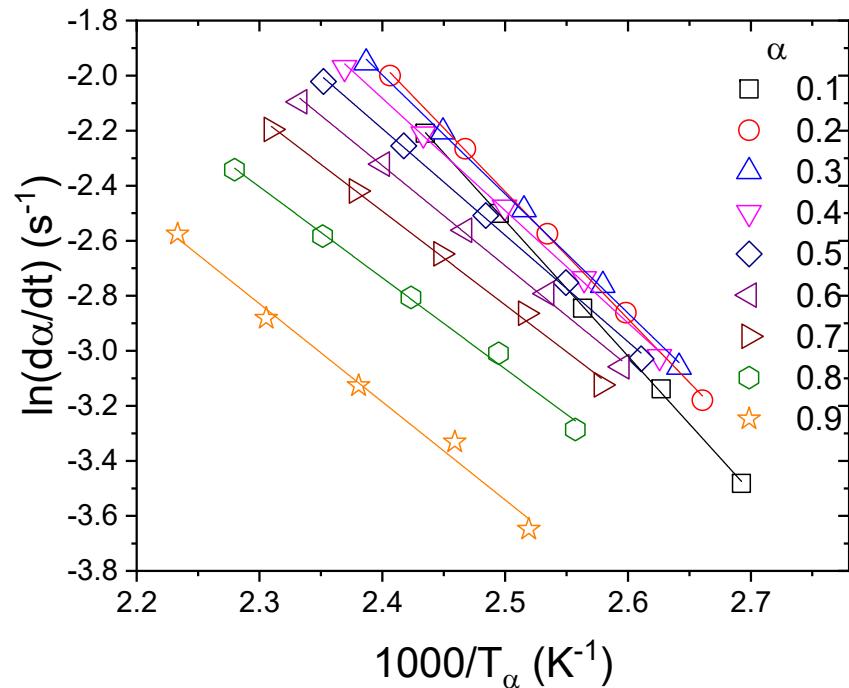


For a given α linear relationship between $\frac{d\alpha}{dt}$ and $\frac{1}{T_\alpha}$

$$\ln\left(\frac{d\alpha}{dt}\right)_\alpha = -\frac{E_\alpha}{RT_\alpha} + \ln(Af(\alpha))$$

Friedman differential isoconversional method.

For each β_i , $d\alpha/dt|_{\alpha,i}$ and $T_{\alpha,i}$ are determined



$$\ln\left(\frac{d\alpha}{dt}\right)_{\alpha,i} = -\frac{E_\alpha}{RT_{\alpha,i}} + \ln(Af(\alpha))$$

Vyazovkin's advanced integral isoconversional method,
minimum of function $\Omega(E_\alpha)$:

$$\Omega(E_\alpha) = \sum_{i=1}^n \sum_{j=1, j \neq i}^n \frac{\beta_j \Delta p(x_{\alpha,i})}{\beta_i \Delta p(x_{\alpha,j})}$$

$$\Delta p(x_\alpha) \equiv p(x_\alpha) - p(x_{\alpha-\Delta\alpha}) = \int_{x_\alpha}^{x_{\alpha-\Delta\alpha}} \frac{e^{-u}}{u^2} du, u = \frac{E}{RT}$$

Once E_α and $Af(\alpha)$ have been determined, the evolution of the cure reaction can be obtained from

$$\frac{d\alpha}{dt} = Af(\alpha) e^{-E_\alpha/RT}$$

Prediction for an arbitrary temperature program, $T(t)$.

Discretized at fixed time intervals Δt , $t_k = k \cdot \Delta t$: $T_k = T(t_k)$.

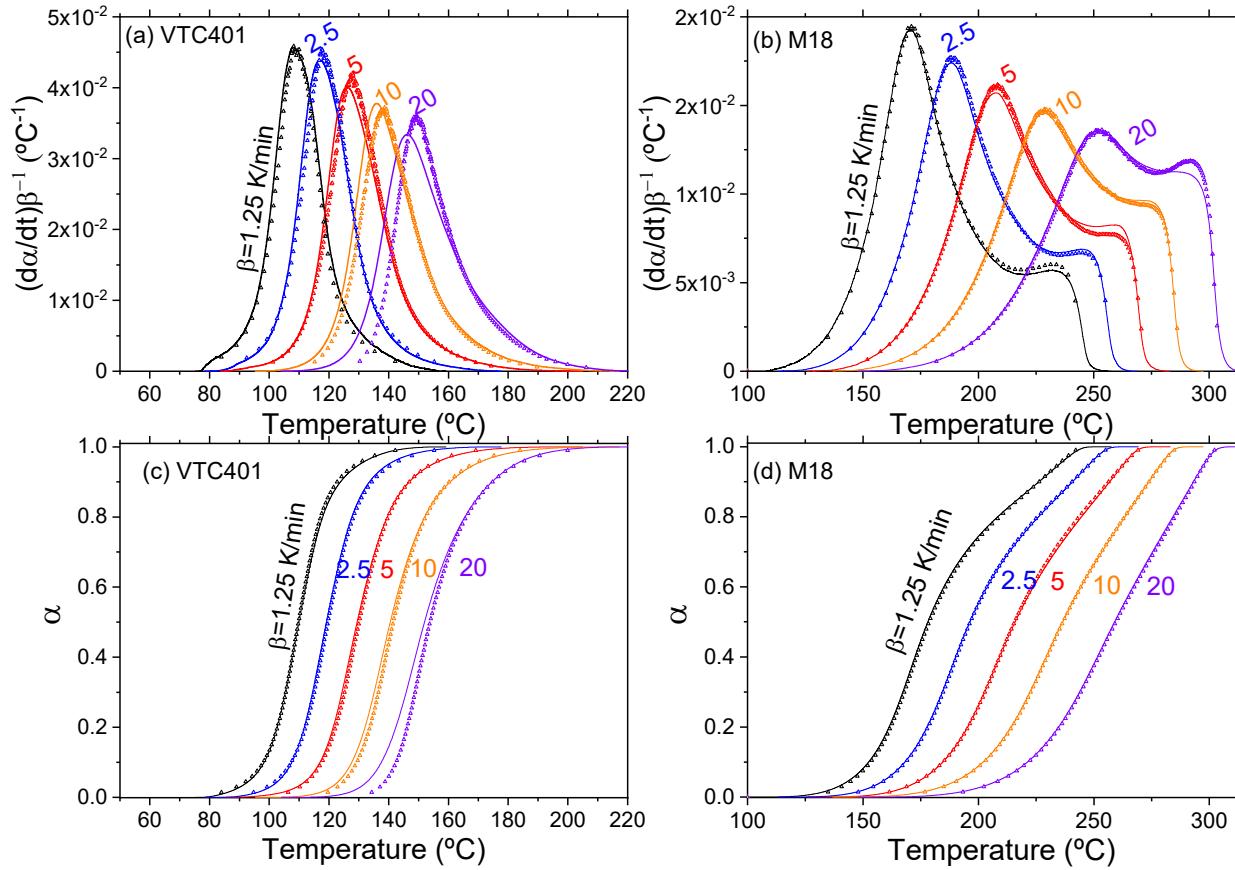
α_j at a given temperature T_j

$$\alpha_{k+1} = \alpha_k + Af(\alpha) \Big|_k \cdot e^{-\frac{E_k}{RT_k}} \Delta t \quad , t_0 = 0 \text{ at } \alpha_0 = 0$$

t_j to reach α_j

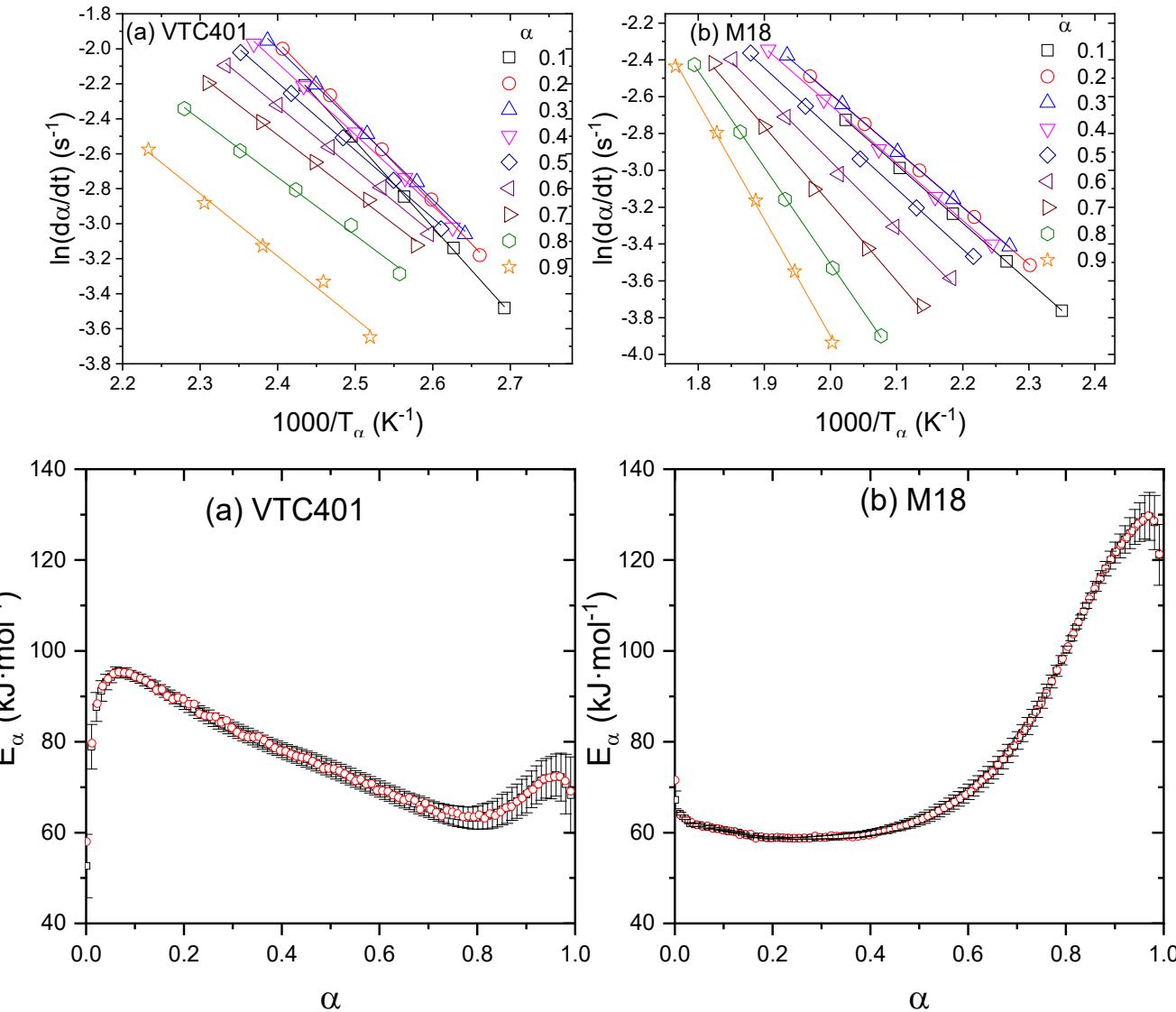
$$t_j = t_{k_j} + \left\{ \frac{1}{n_{exp}} \sum_{i=1}^{n_{exp}} \frac{E_j}{R\beta_i} \left[p \left(\frac{E_j}{RT_{j,i}} \right) - p \left(\frac{E_j}{RT_{j-1,i}} \right) \right] - \left[e^{-\frac{E_j}{RT_{k_{j-1}}}} (t_{k_{j-1}+1} - t_{j-1}) + \sum_{k=k_{j-1}+2}^{k_j} e^{-\frac{E_j}{RT_k}} \Delta t \right] \right\} e^{-\frac{E_j}{RT_k}}$$

DSC measurements

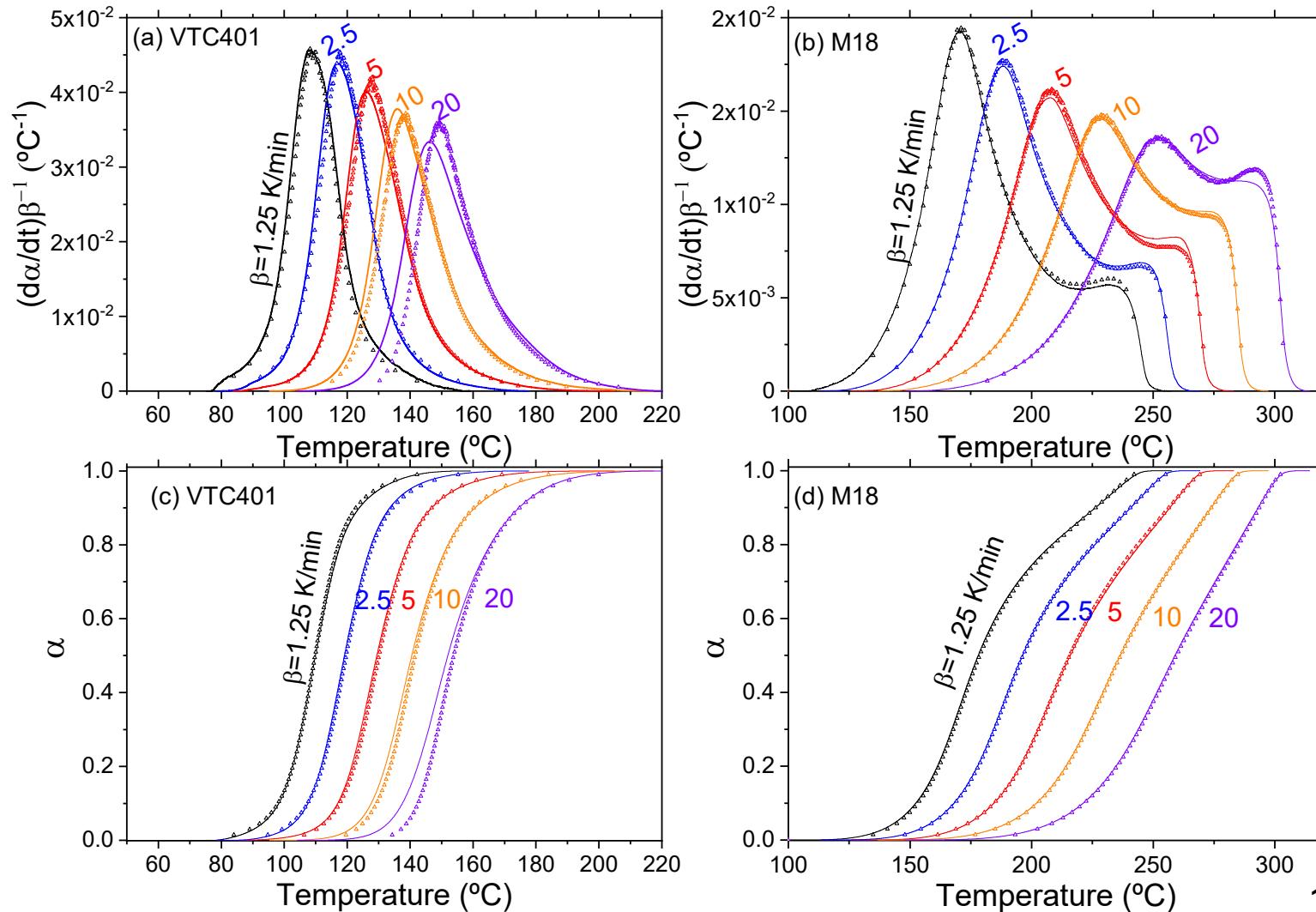


| β (K/min) | ΔH (J/g) | |
|-----------------|------------------|-----|
| | VTC401 | M18 |
| 1.25 | 687 | 688 |
| 2.5 | 684 | 683 |
| 5 | 662 | 667 |
| 10 | 641 | 695 |
| 20 | 647 | 677 |

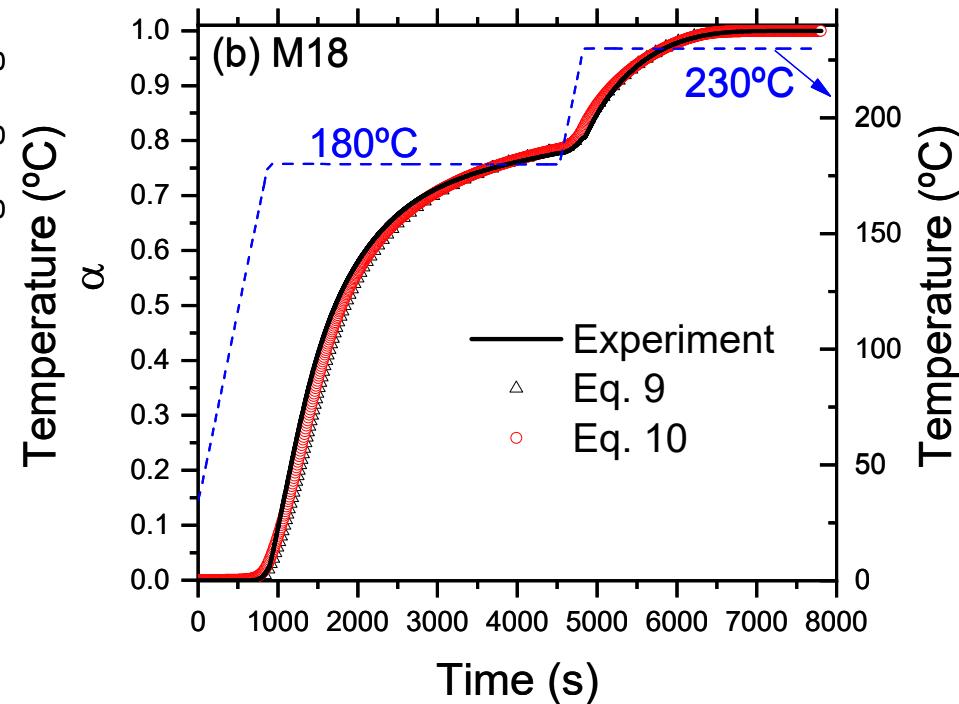
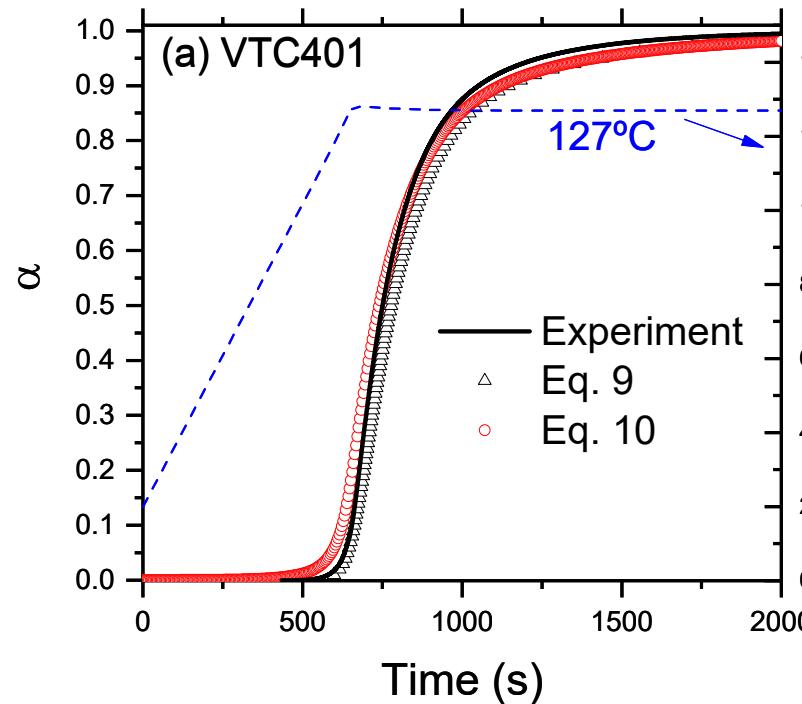
Friedman kinetic analysis



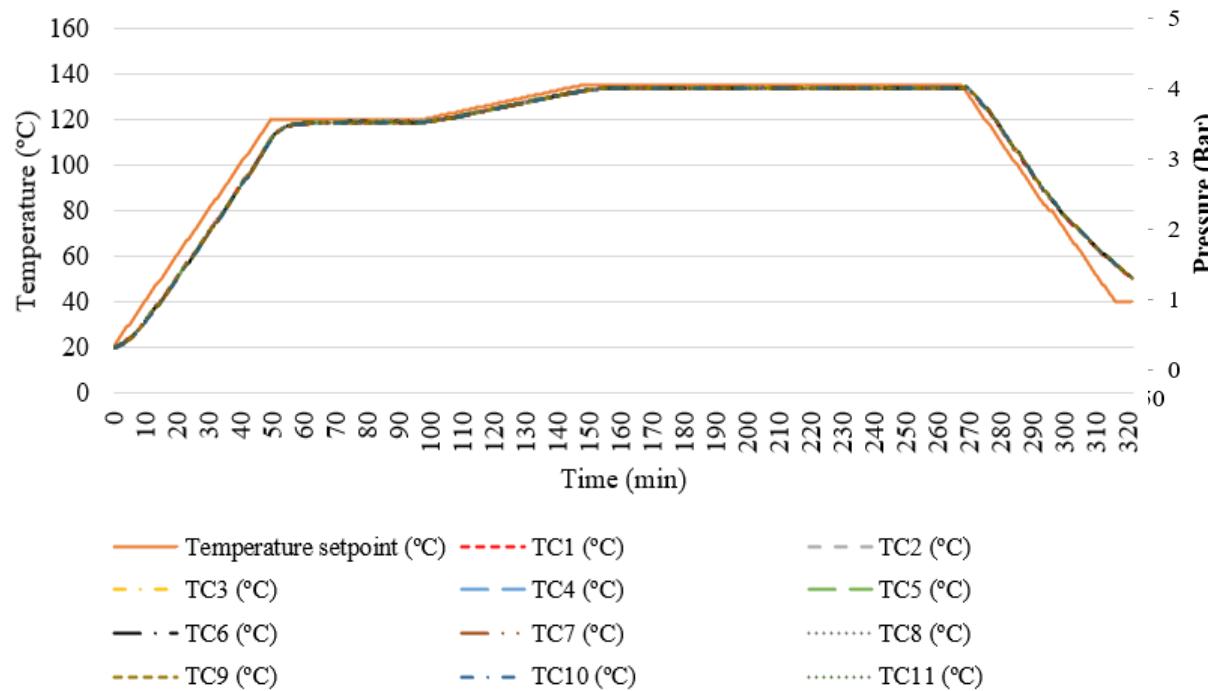
Self consistency test



Prediction test

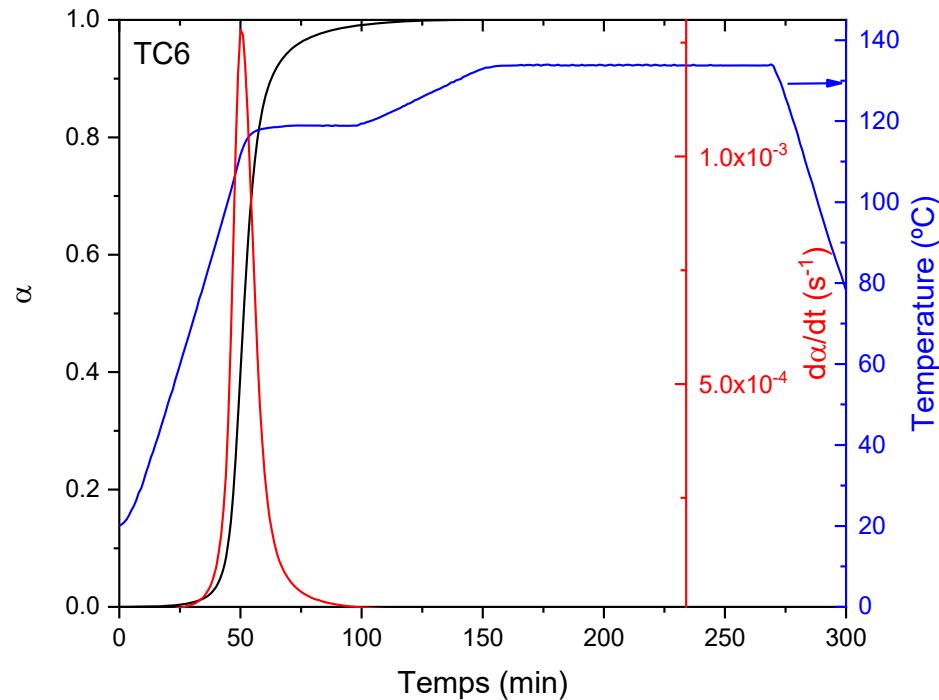


More than 5 hours, is it possible to reduce the time
needed to fully cure the resin?

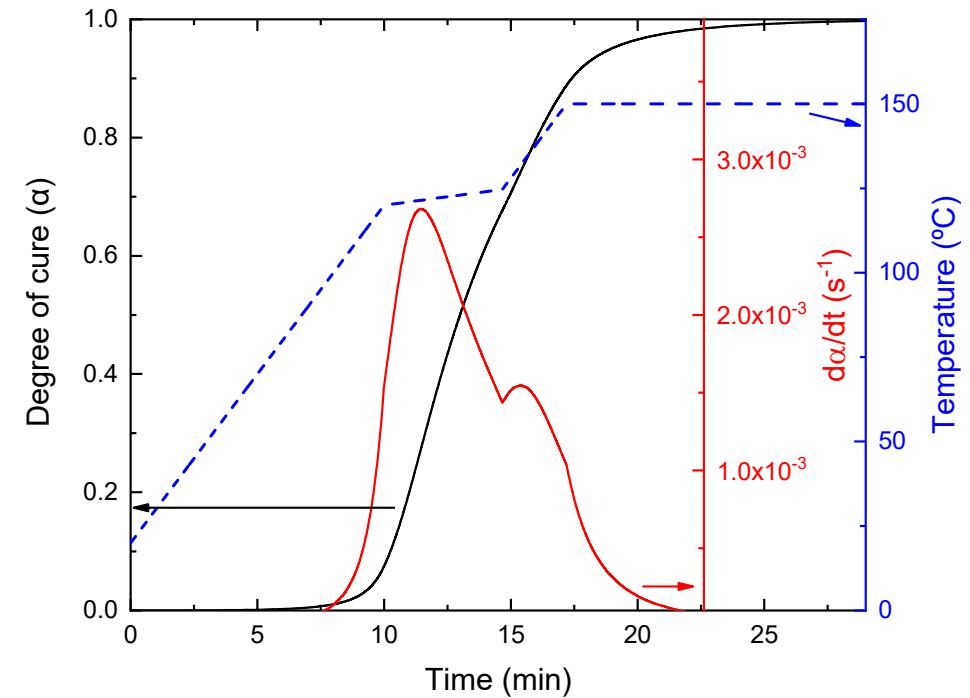


Manufacturer's recommended cure cycle

Predicted evolution:

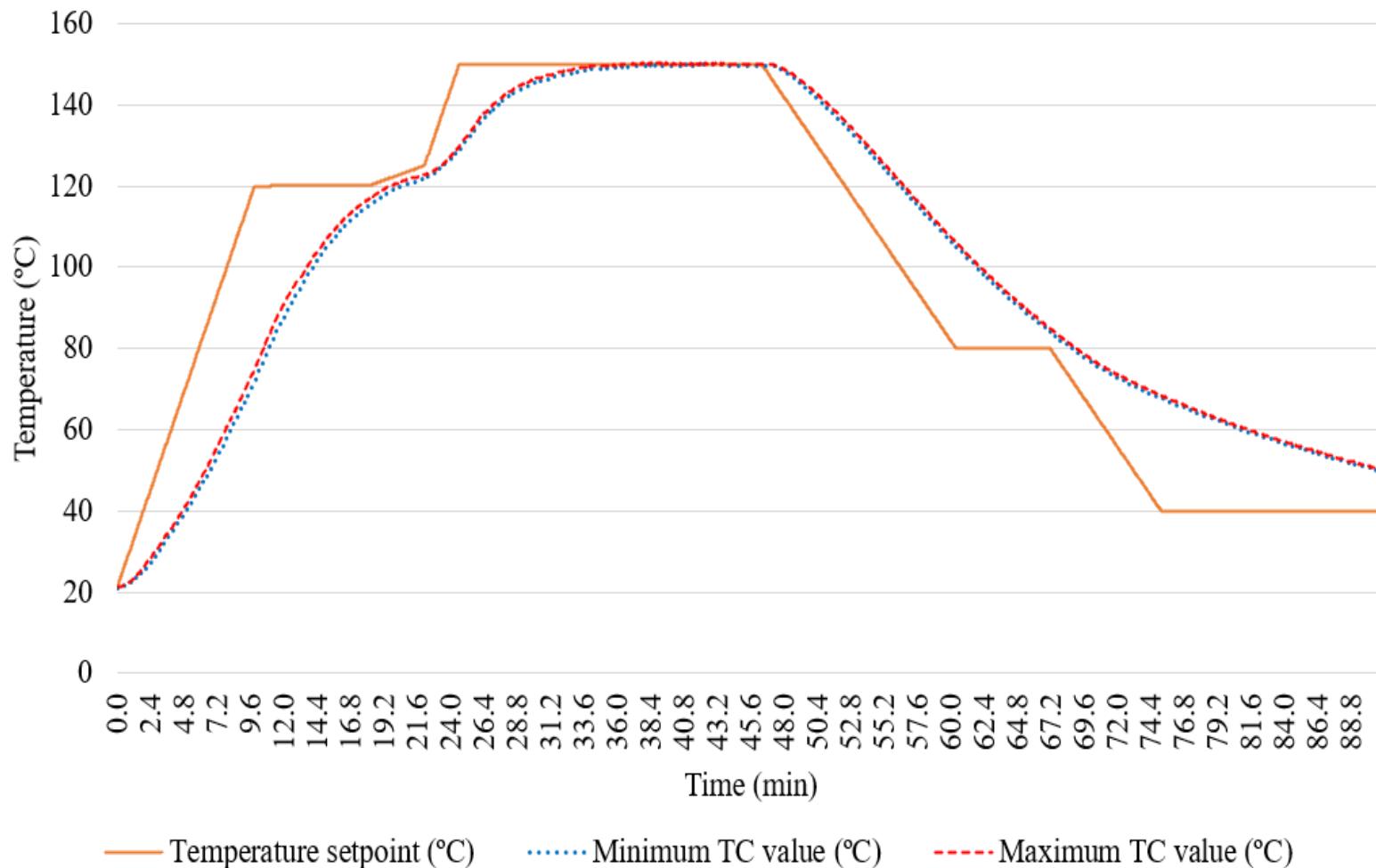


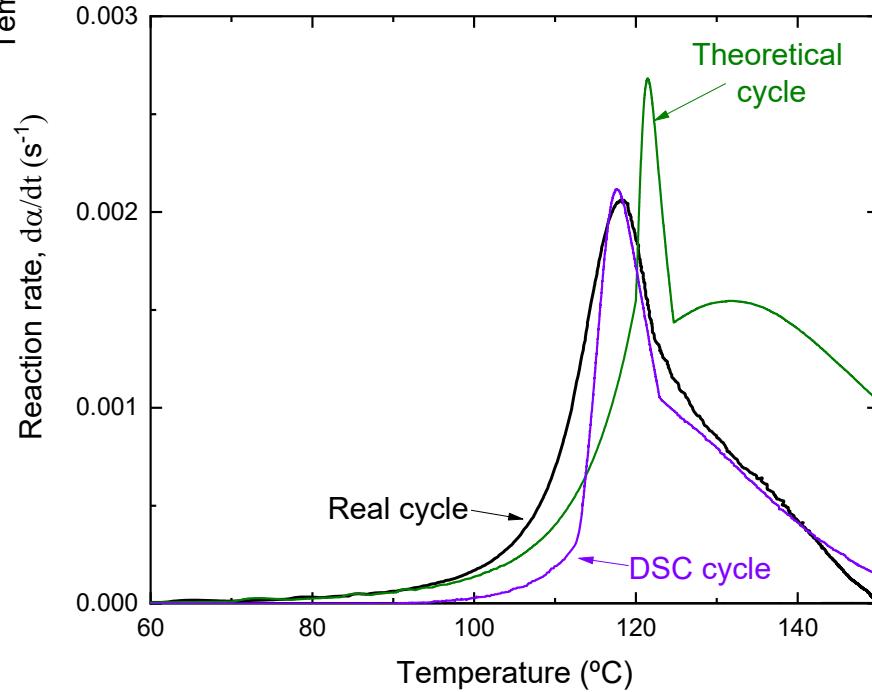
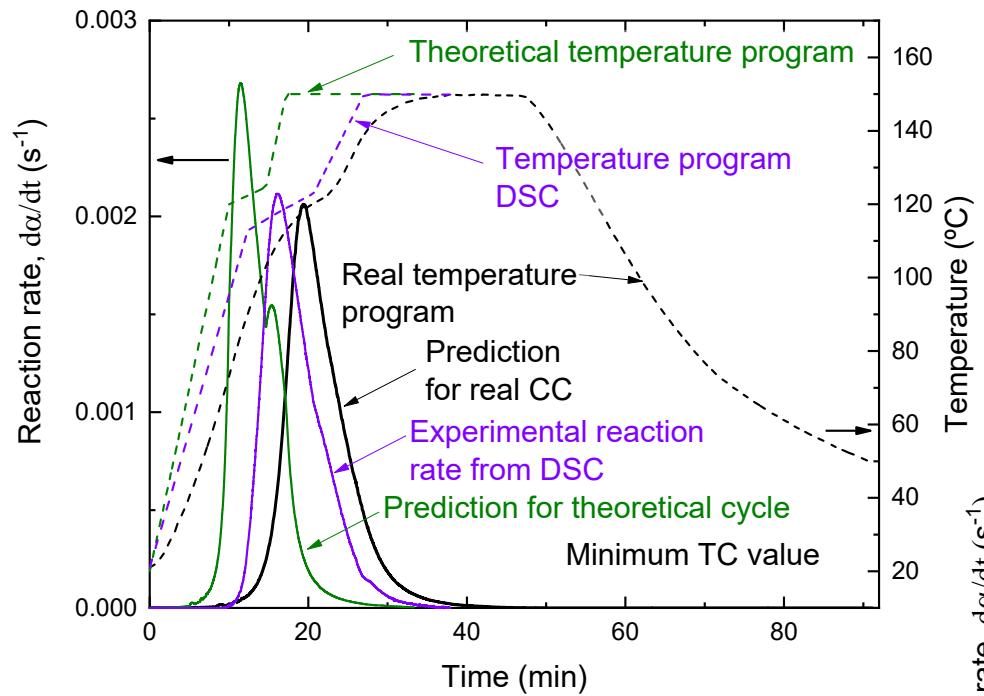
Manufacturer's recommended
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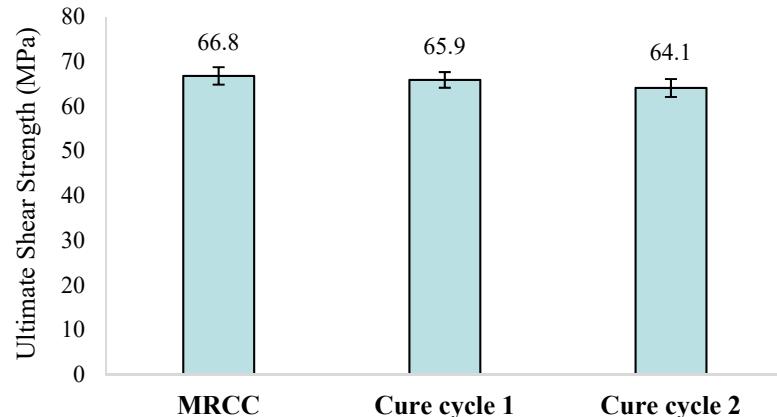
Optimized
cure cycle

Real cure cycle

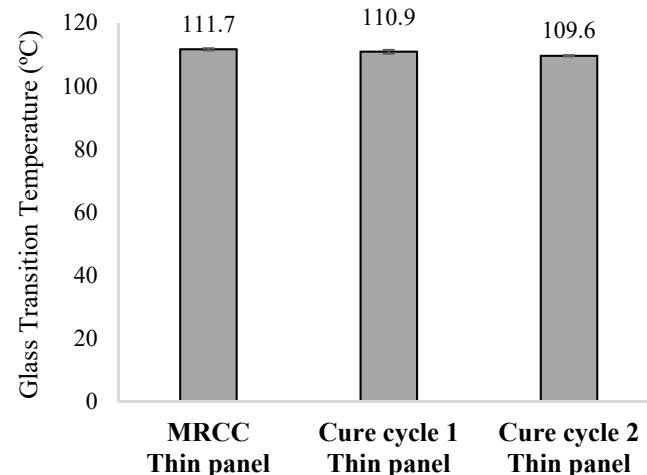
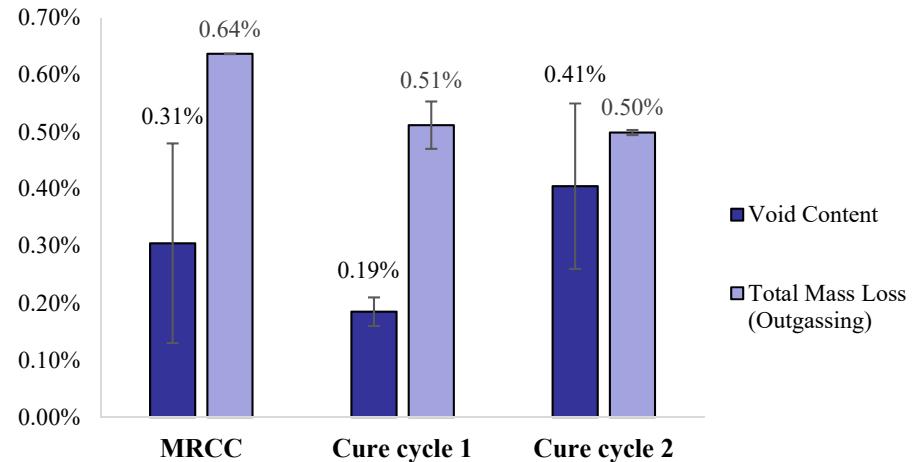


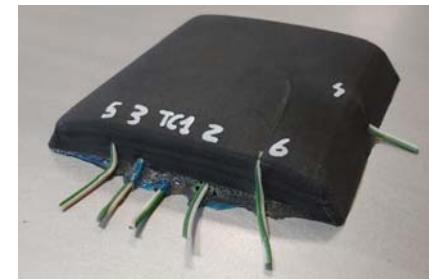
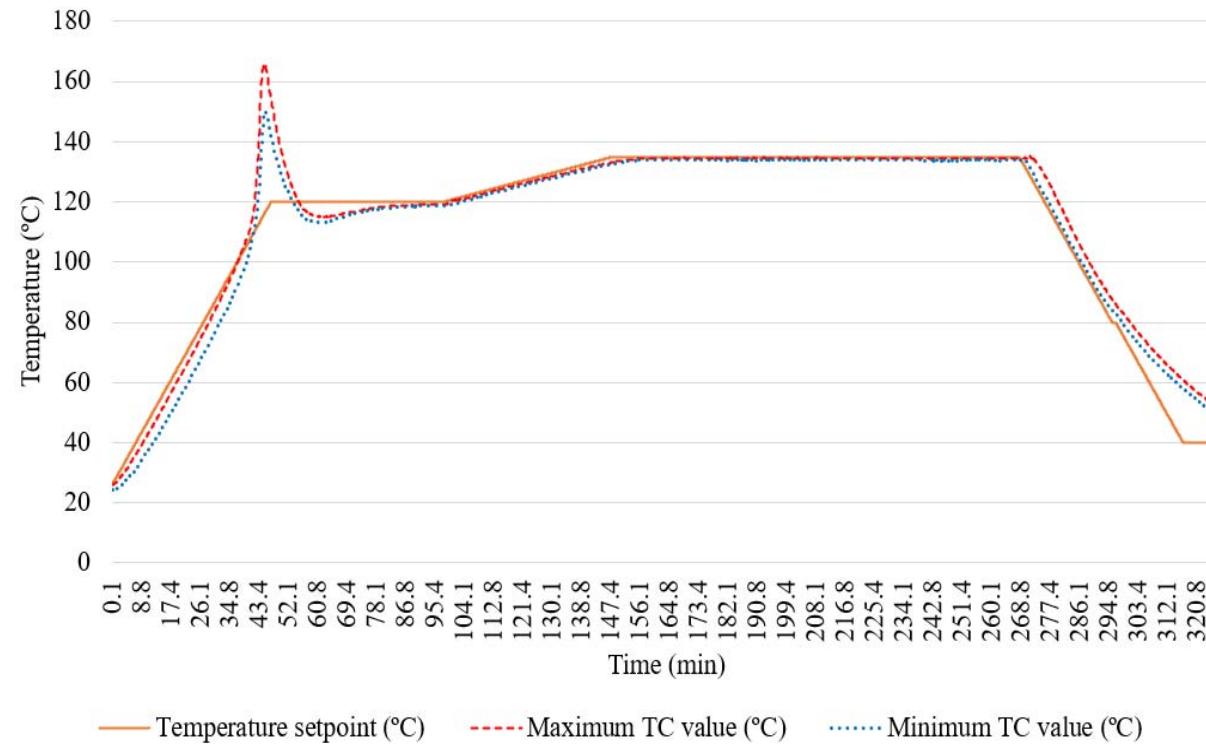


Mechanical properties

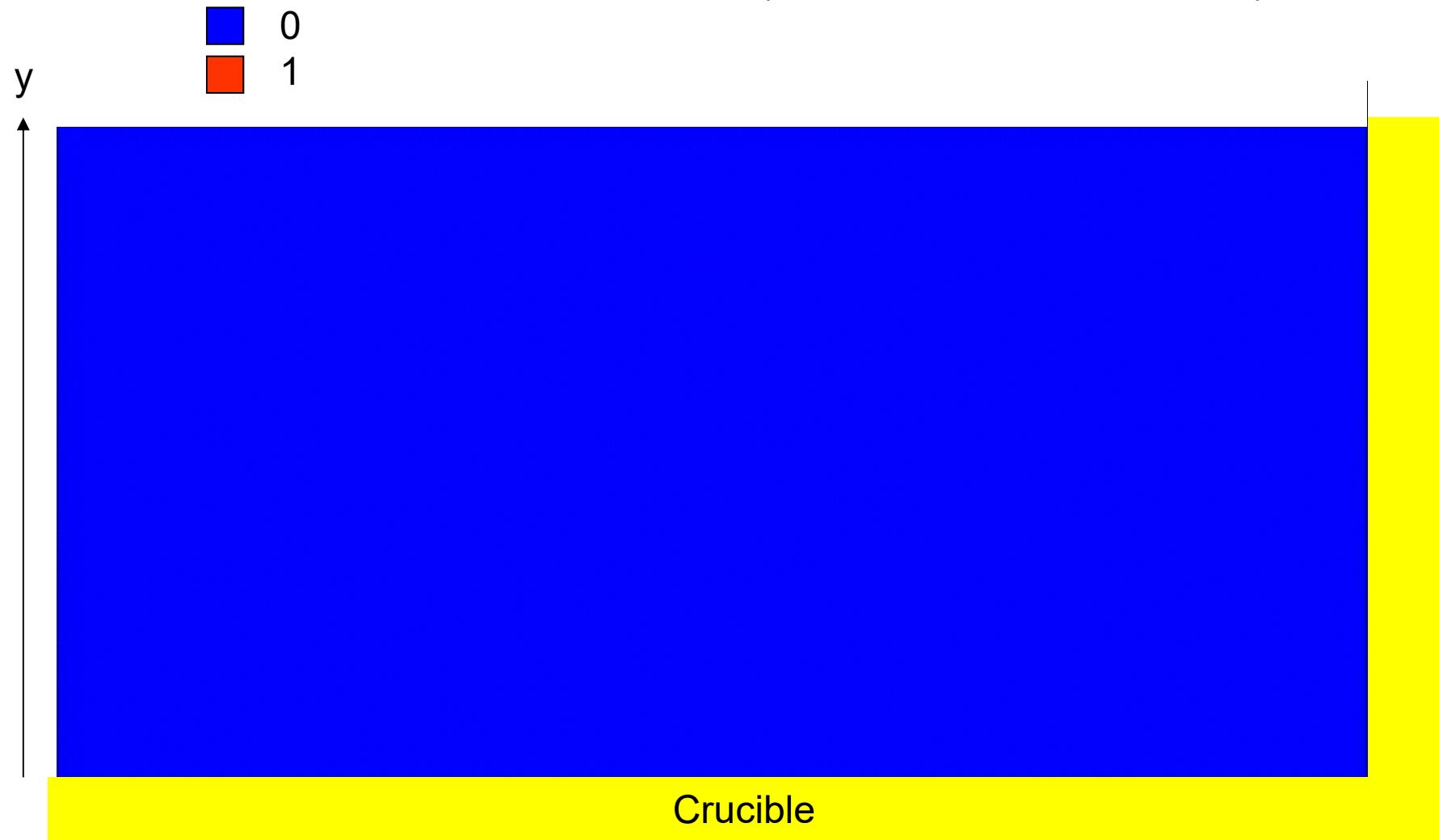


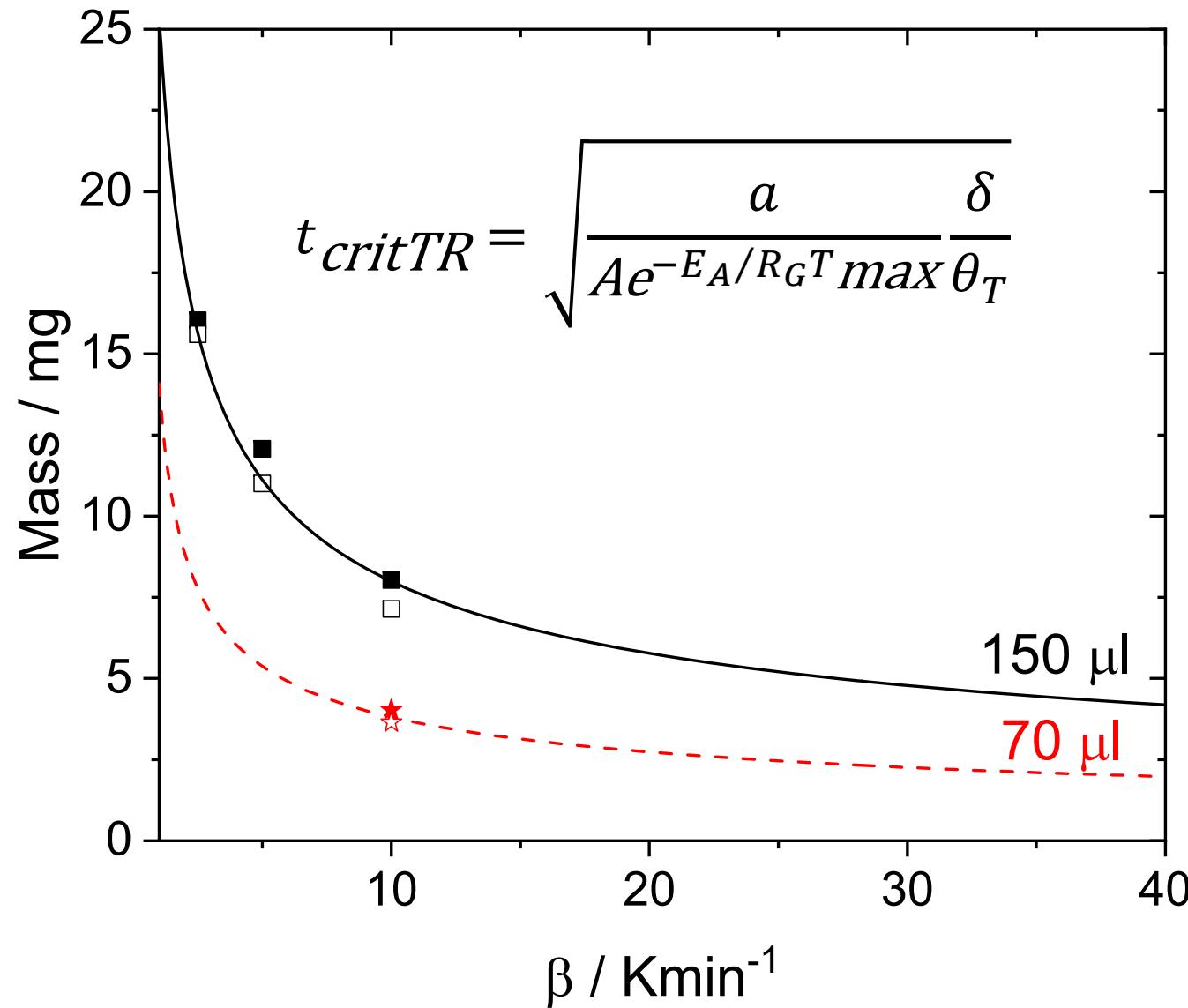
Physical properties





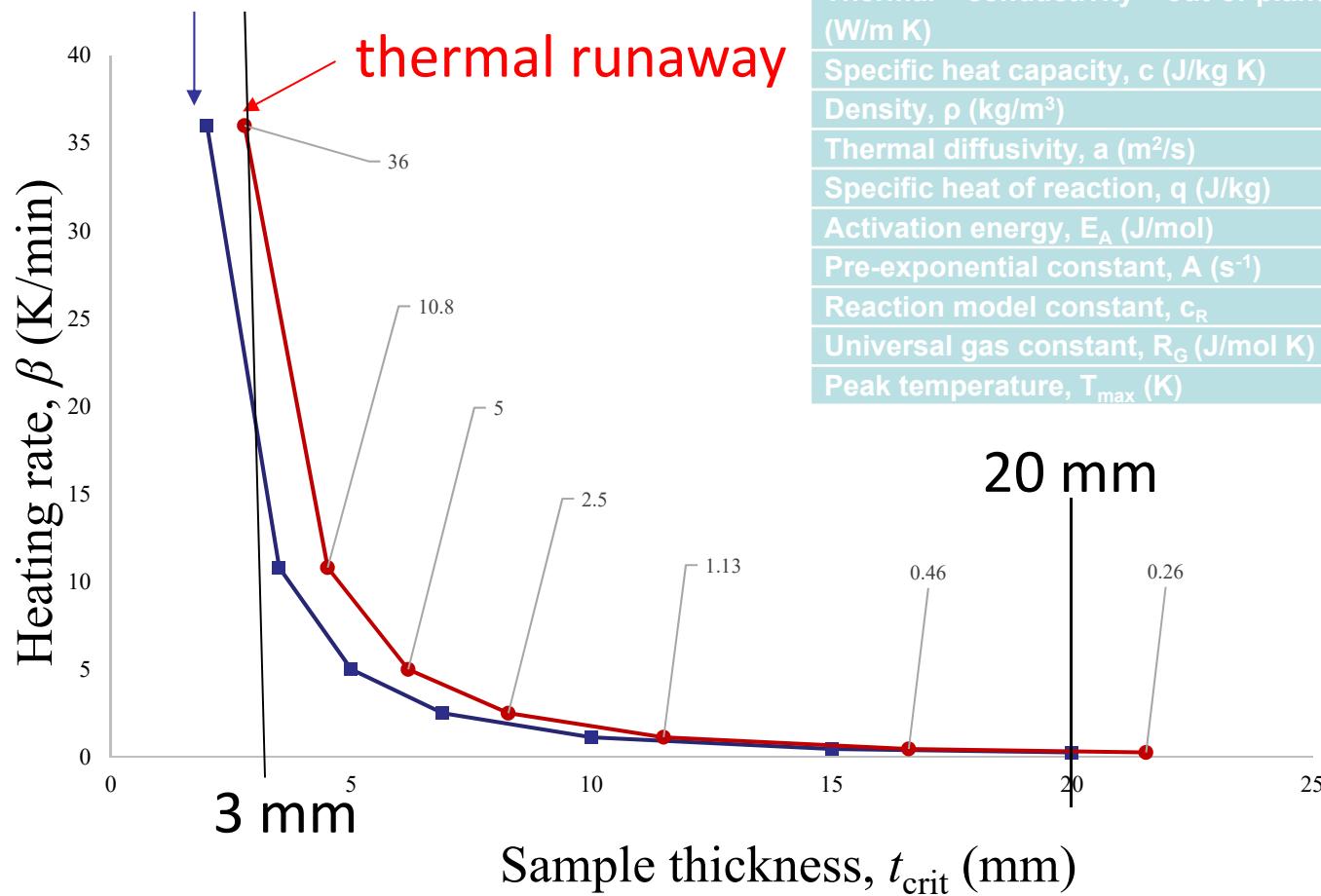
Evolution of the transformed fraction (constant transformation rate):





$$t_{crit} = 1.5 \left(10^{-5} \Delta T \frac{caE_A}{R_G q} \frac{c_R}{A e^{-E_A/R_G T_{max}}} \right)^{1/2}$$

10 °C overheating



Conclusions

- During the curing reaction there is a dramatic change of the physical properties of the material.
- The kinetics of the curing process is usually complex and involves more than one process. At the beginning is controlled by the chemical reaction but at the end diffusion is the kinetics limiting process.
- Isoconversional methods allow to fit the kinetics and predict the evolution.
- A total reduction in real time of up to 74 % has been achieved.
- It is possible to analytically determine the critical thickness for overheating.

Thermal decomposition of calcium propionate: films and powders



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